

# Mechanical Degradation of Tungsten Alloys at Extreme Temperatures in Vacuum and Oxidation Atmospheres



EFDA

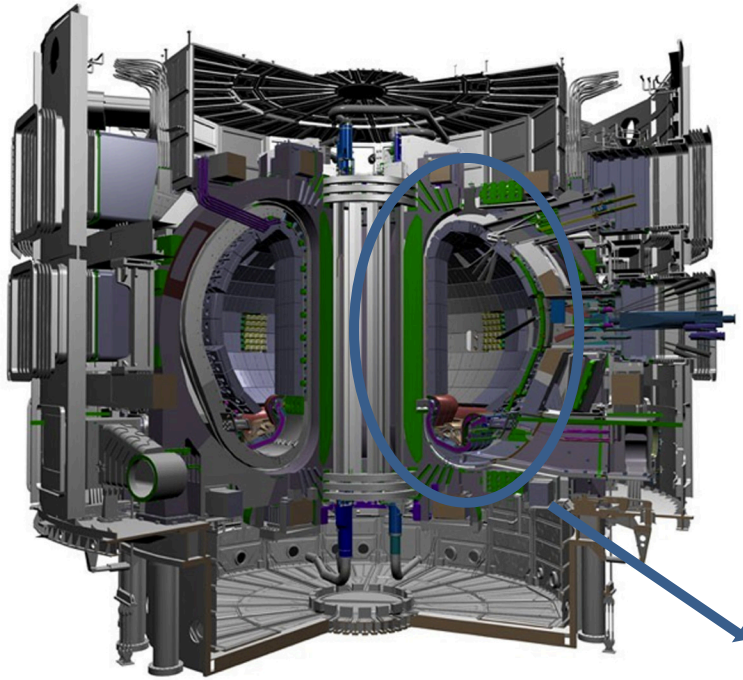
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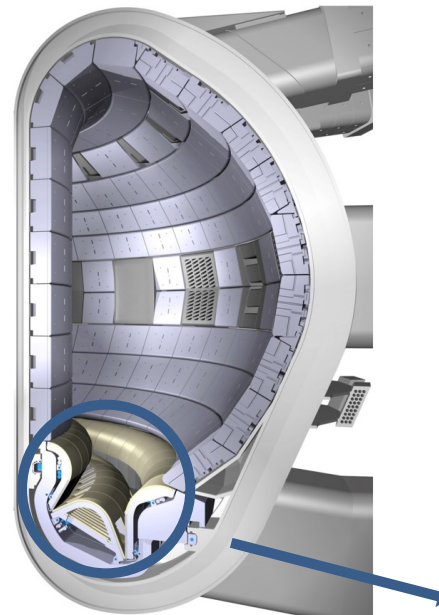
## motivation



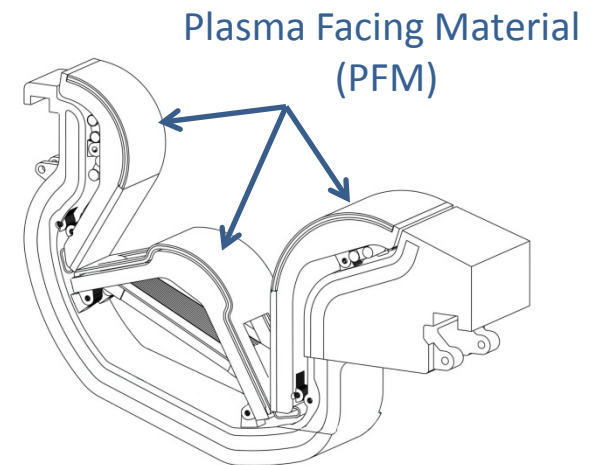
fusion reactor

### Extreme conditions

- High ion/neutron flux
- High heat load ( $\sim 10 \text{ MW/m}^2$ )
- High temperature
- Thermal stresses & cycling
- Off normal events (e.g. plasma disruption)



vacuum vessel



divertor

## why tungsten as PFM?

features an **unique property combination**

- the highest melting point of all metals
- good thermal conductivity
- low tritium retention
- low physical sputtering yield
- high thermal resistance
- high temperature strength



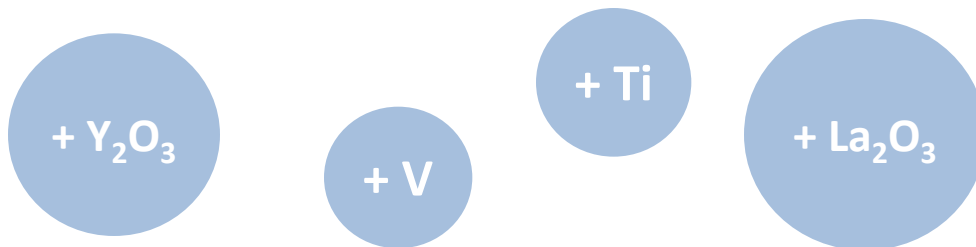
at the divertor perform also a structural application, so **ductility is** needed and W has a **brittleness** problem  
 oxidation resistance



Operation temperature range

- Lower limit: ductile-brittle transition temperature (DBTT) 400-650 ° C
- Upper limit: rxx temperature (RT) 1300 ° C

rxx processes and irradiation effects during operation can induce embrittlement



W – 1 wt%  $Y_2O_3$  (W-1Y)  
 W – 2 wt% Ti – 1 wt%  $La_2O_3$   
 (W-2T1L)

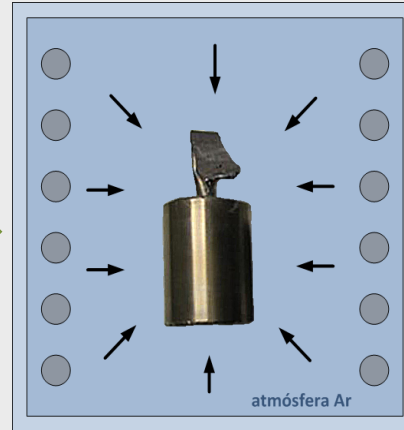
## material processed

<b>W</b>	99.9 %	< 5 $\mu\text{m}$
<b>Ti</b>	99.9 %	< 110 $\mu\text{m}$
<b>La<sub>2</sub>O<sub>3</sub></b>	99.5 %	0.01-0.05 $\mu\text{m}$
<b>Y<sub>2</sub>O<sub>3</sub></b>	99.5 % mix, 4 h	0.01-0.05 $\mu\text{m}$

MA  
20h



canned  
degassed  
400 ° C, 24 h



HIP

1300 ° C, 2 h, 195 MPa produced in UC3M, Spain

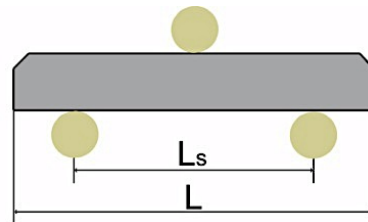


$\varnothing = 30 \text{ mm}$   
 $L = 50 \text{ mm}$

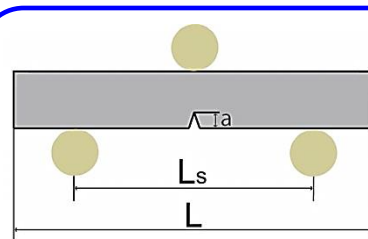
TPB tests samples



1.6 x 1.6 x 25 mm<sup>3</sup>



smooth bend bars



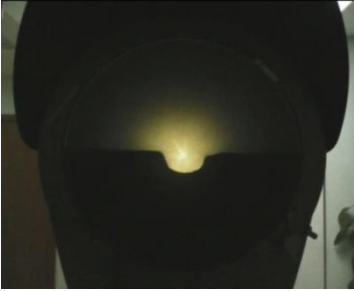
notched bend bars

**HOW?**

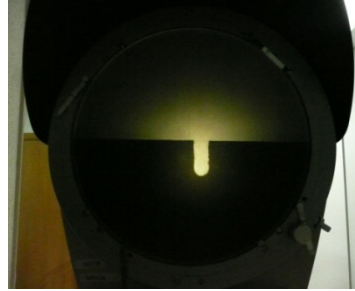
$L = 25 \text{ mm}$   
 $B = 1.6 \text{ mm}$   
 $D = 1.6 \text{ mm}$   
 $L_s = 16 \text{ mm}$   
 $a \sim 400 \mu\text{m}$



## notched samples

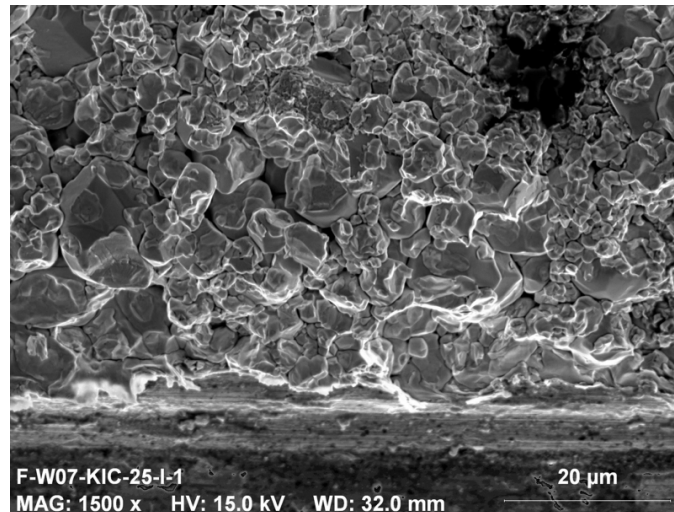
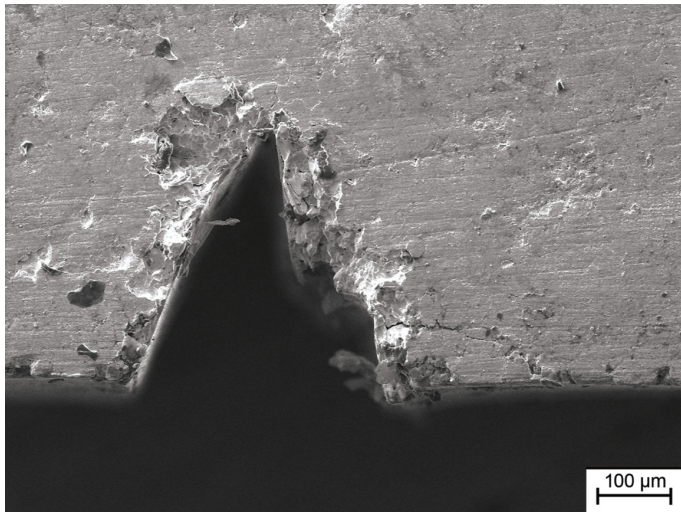
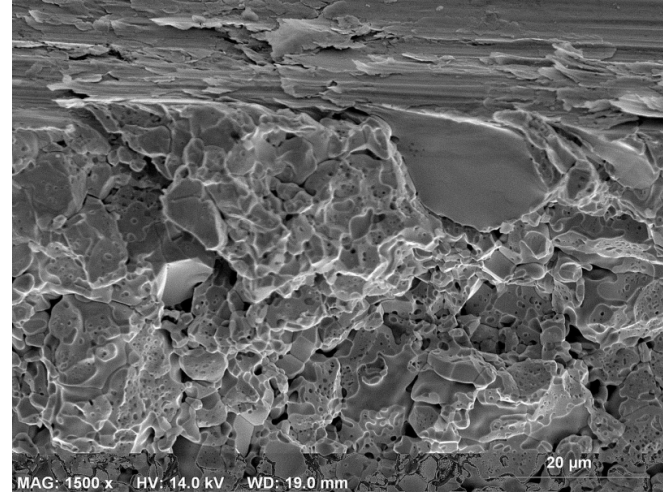


diamond disc  
~ 200  $\mu\text{m}$



diamond wire  
~ 75  $\mu\text{m}$

fast, plastic damage, big notch root radius



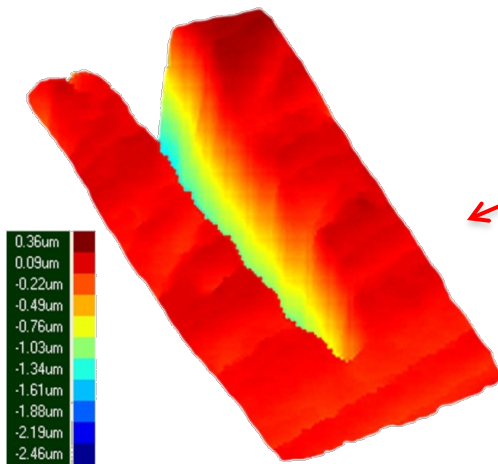
metal blade  
~ 5-7  $\mu\text{m}$

- no plastic damage
- slow complex process

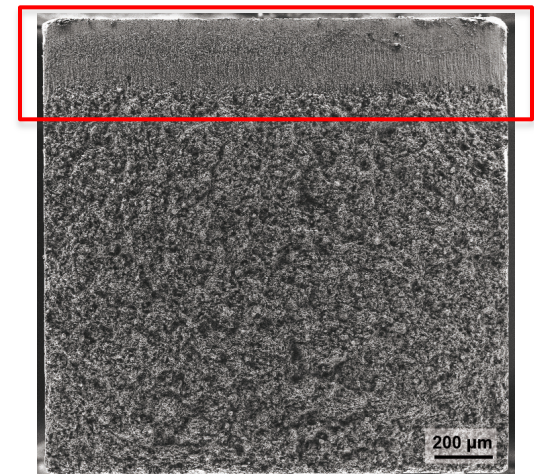
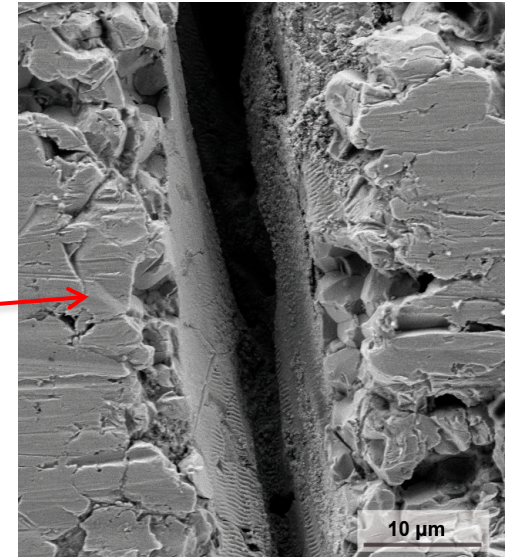
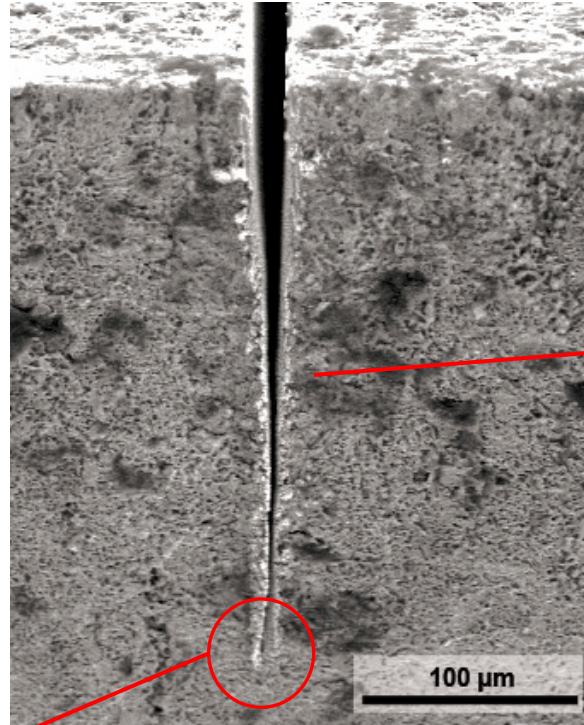
## notched samples

laser  
1-50 nm

- no plastic damage
- no rxn of the tip grains
- similar to a crack



Profilometer image



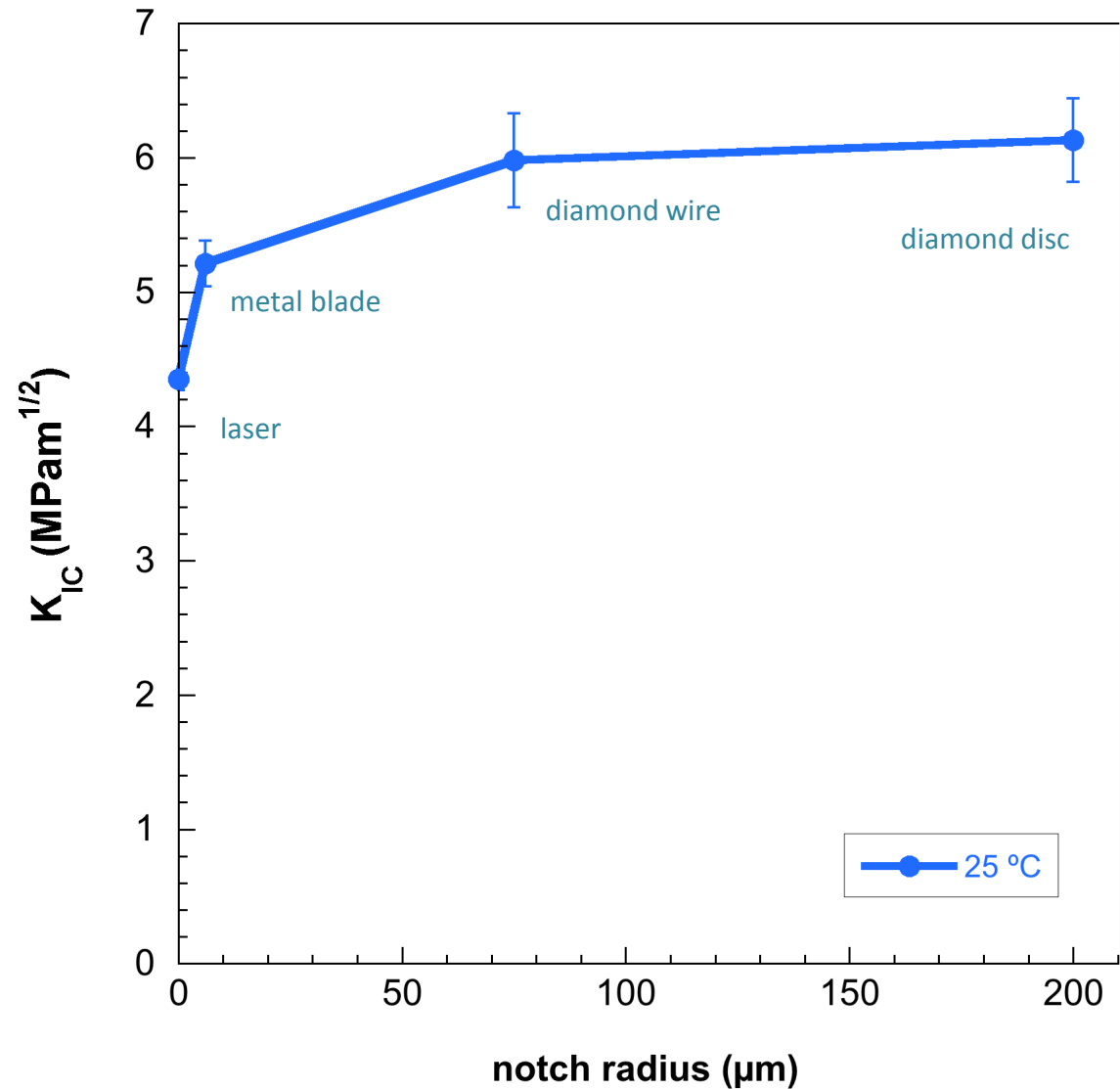


## notched samples

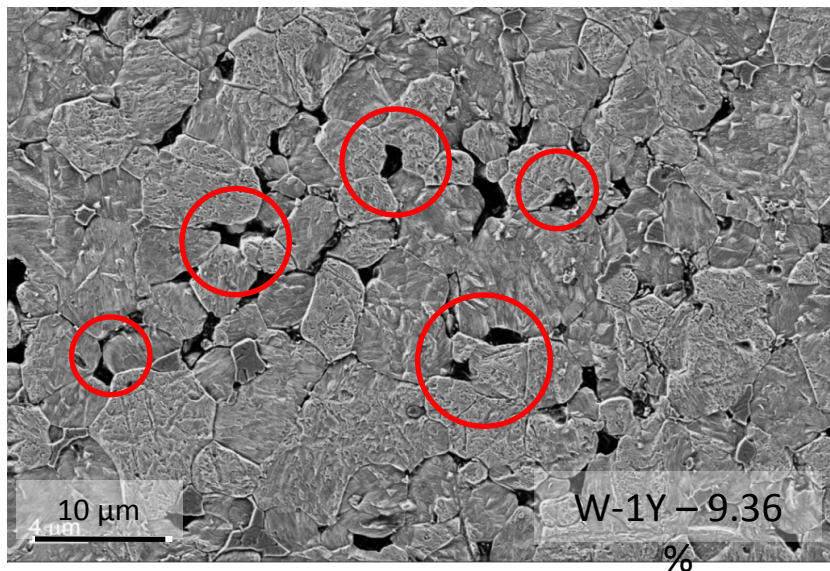
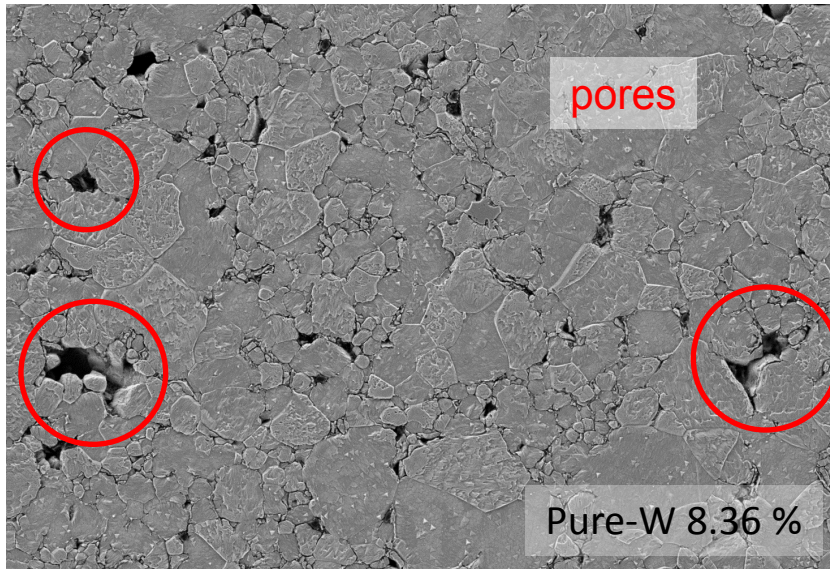
TPB tests were performed in notched samples of the 4 methods to check the notch root radius effect



↓ dispersion results  
 ↓ 25-30% of the value

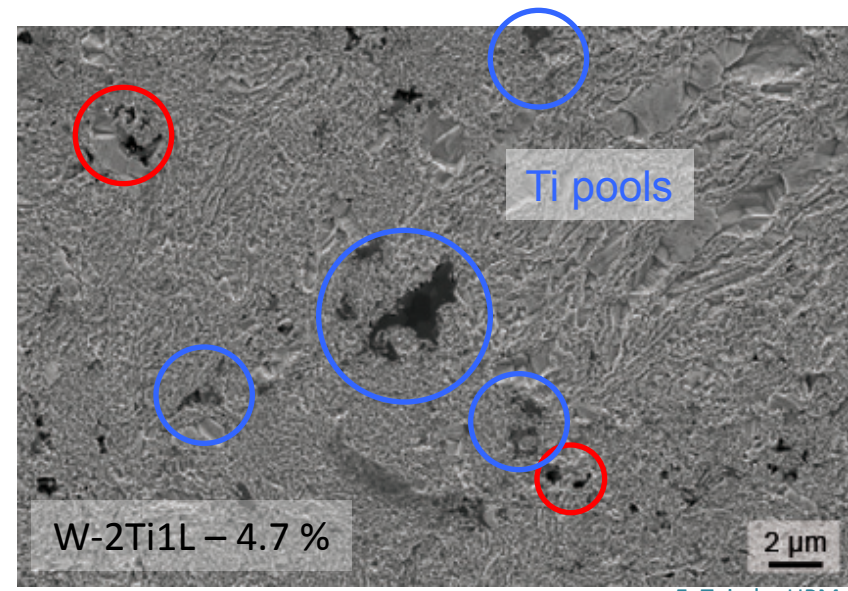


## microstructure



### W-2Ti1L:

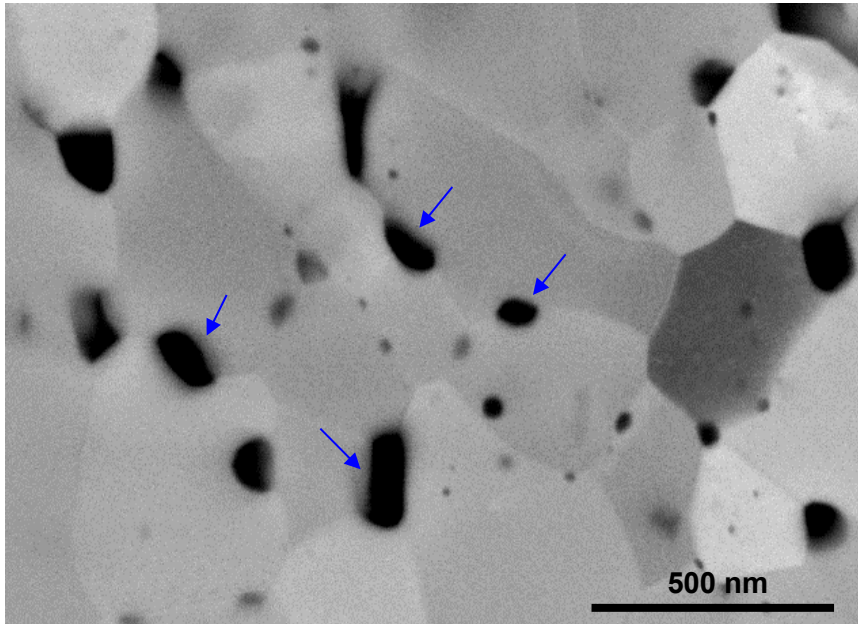
- with the addition of the alloying elements the grain size become nanometric
- change from polyedral coarse grains (pure-W) to coarse W grains and Ti pools surrounded by a W-Ti-La solid solution



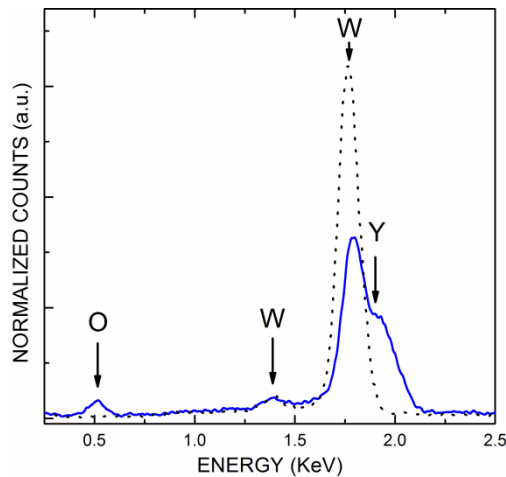
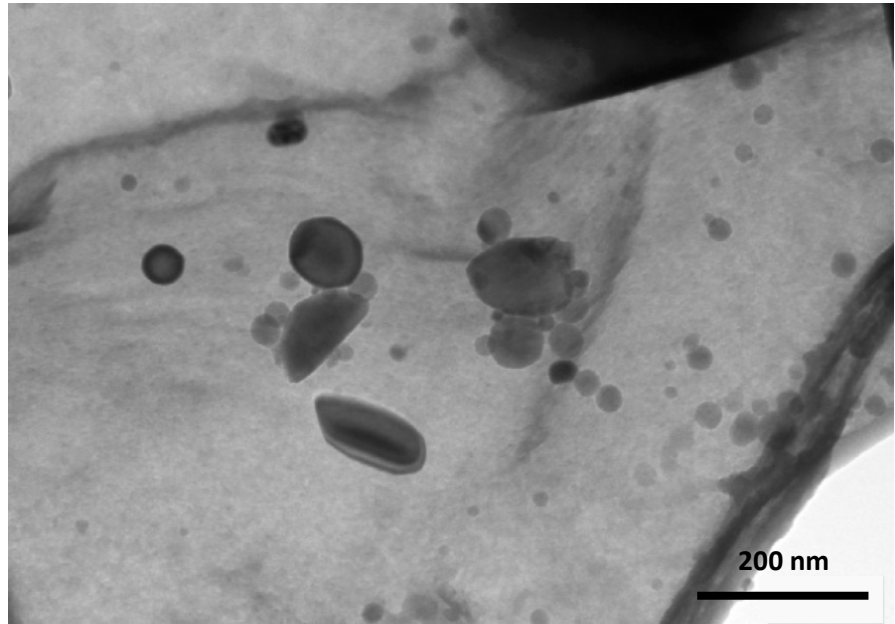
E. Tejado, UPM



## microstructure (W-1Y)



C. Ballesteros, UC3M

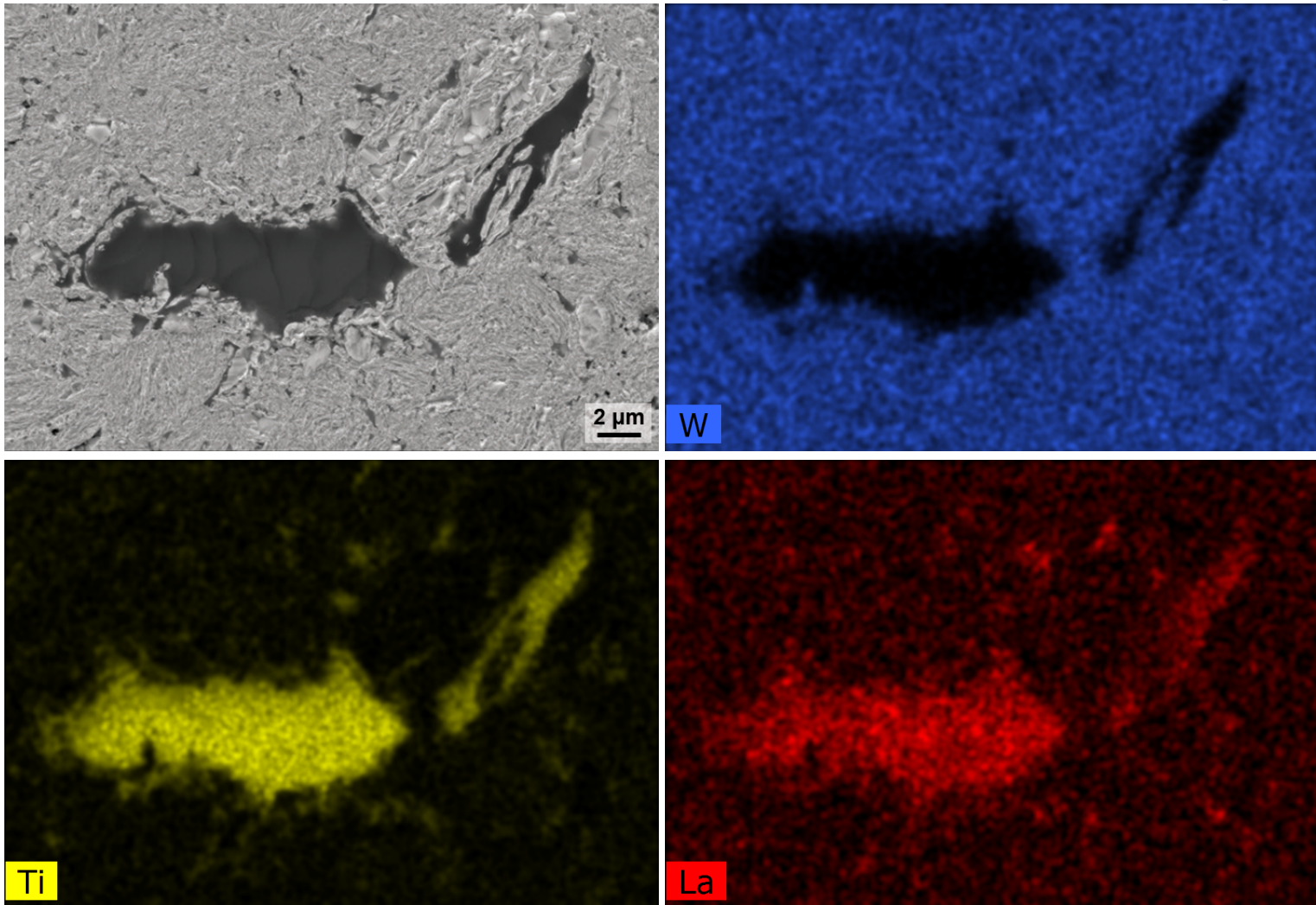


edx spectra from  
black contrast areas

$Y_2O_3$  nanoparticles:

- spheroidal dispersoids with variable size
- located in the grain boundaries
- sometimes agglomeration

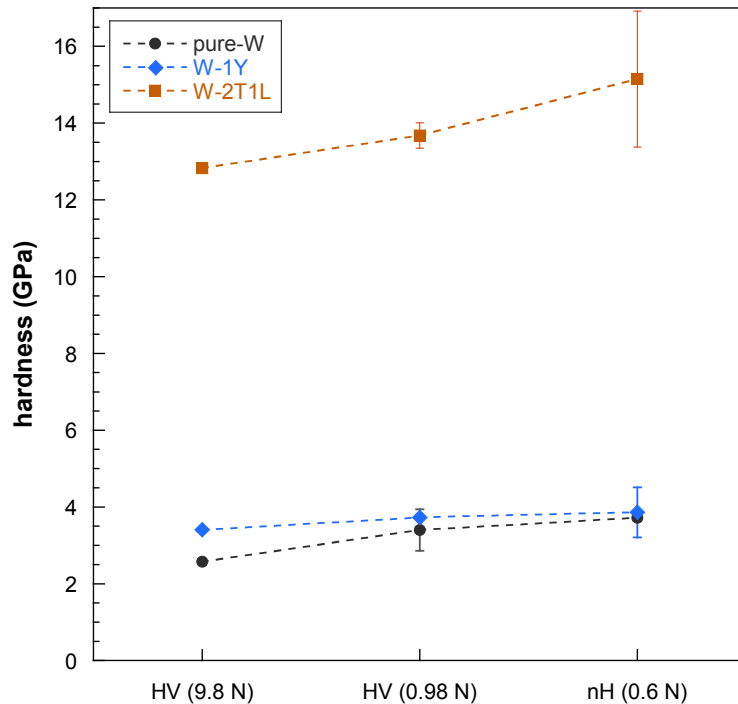
## microstructure (W-2Ti1L)



E. Tejado, UPM

FESEM image of W-2Ti-1La<sub>2</sub>O<sub>3</sub> alloy with mapping (W), mapping (Ti) and mapping (La).

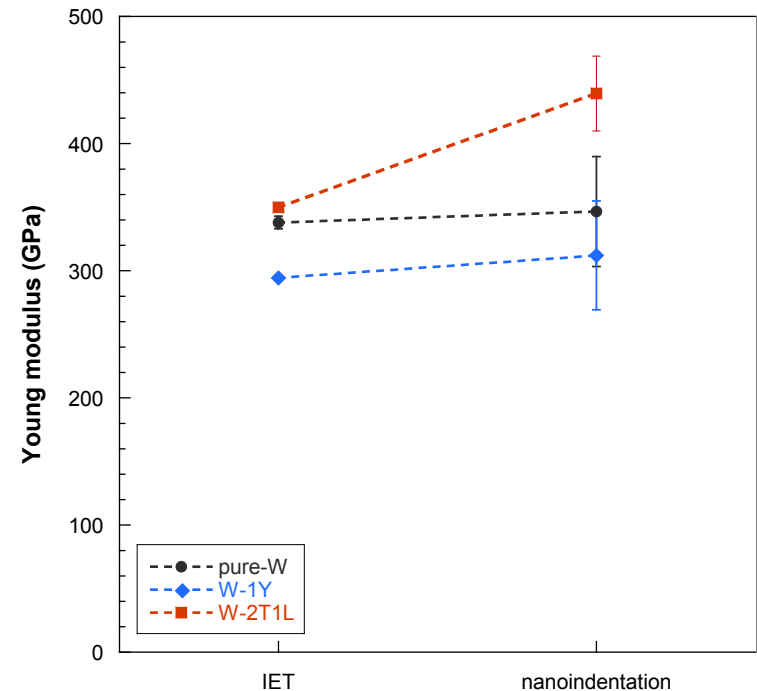
## hardness



small influence of the load  
 comparable results with instrumented  
 method  
 W-2T1L  $\uparrow$  hardness

## mechanical properties

### Young modulus

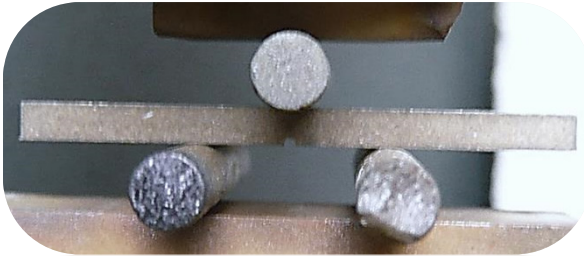


comparable values  
 W-1Y  $\downarrow$  E  
 W-2T-1L Small increase of the values

\*IET= Impulse Excitation Technique

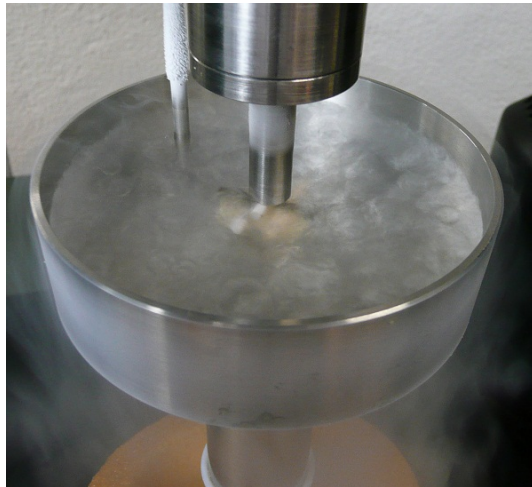


## TPB tests



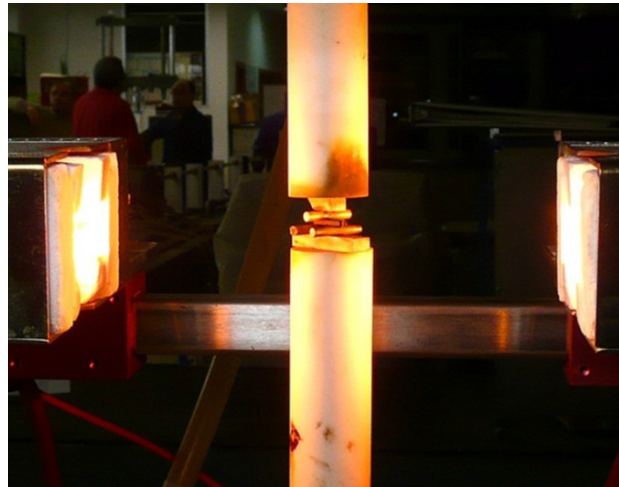
displacement rate:  
 $100 \mu\text{m/min}$   
 heating rate:  
 $50^\circ \text{C/min}$

liquid N immersion



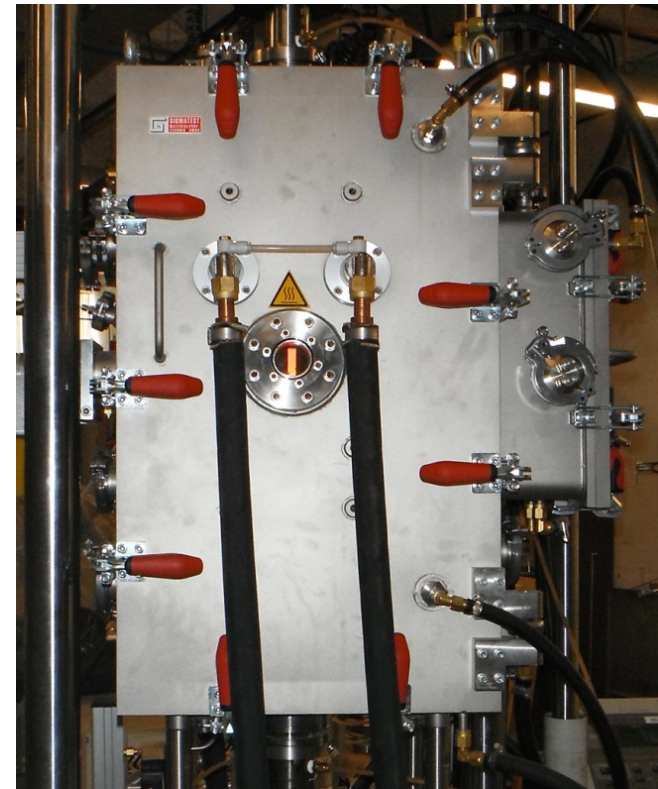
$T = -196^\circ \text{C}$

oxidation atmosphere (air)



$T \leq 1000^\circ \text{C}$

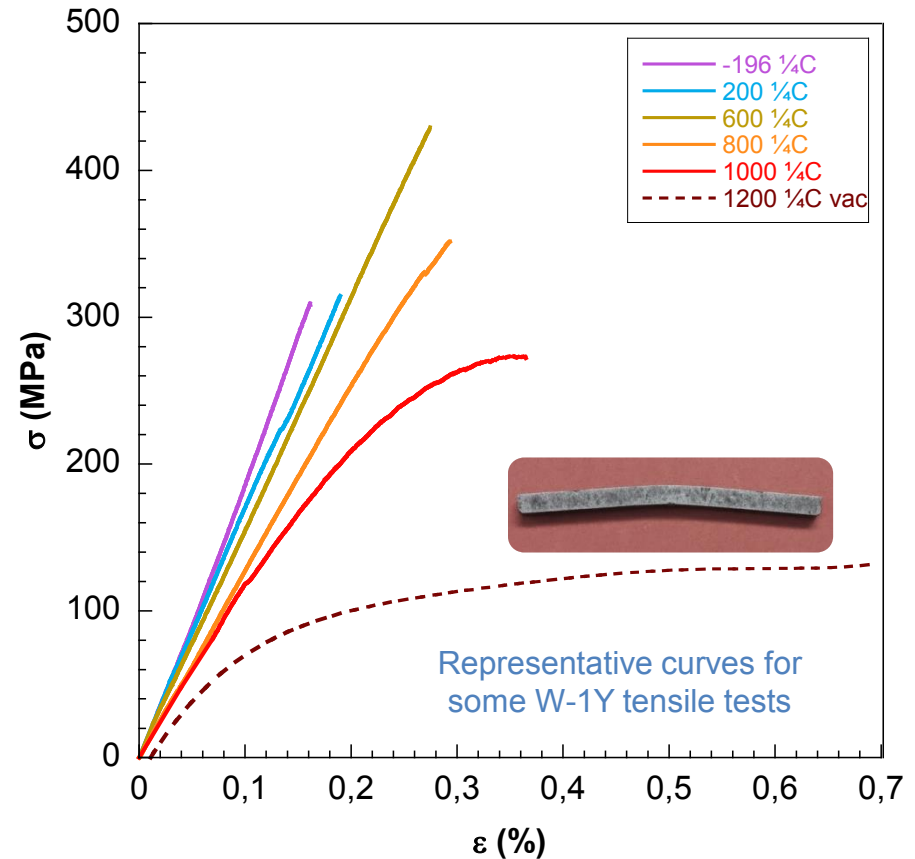
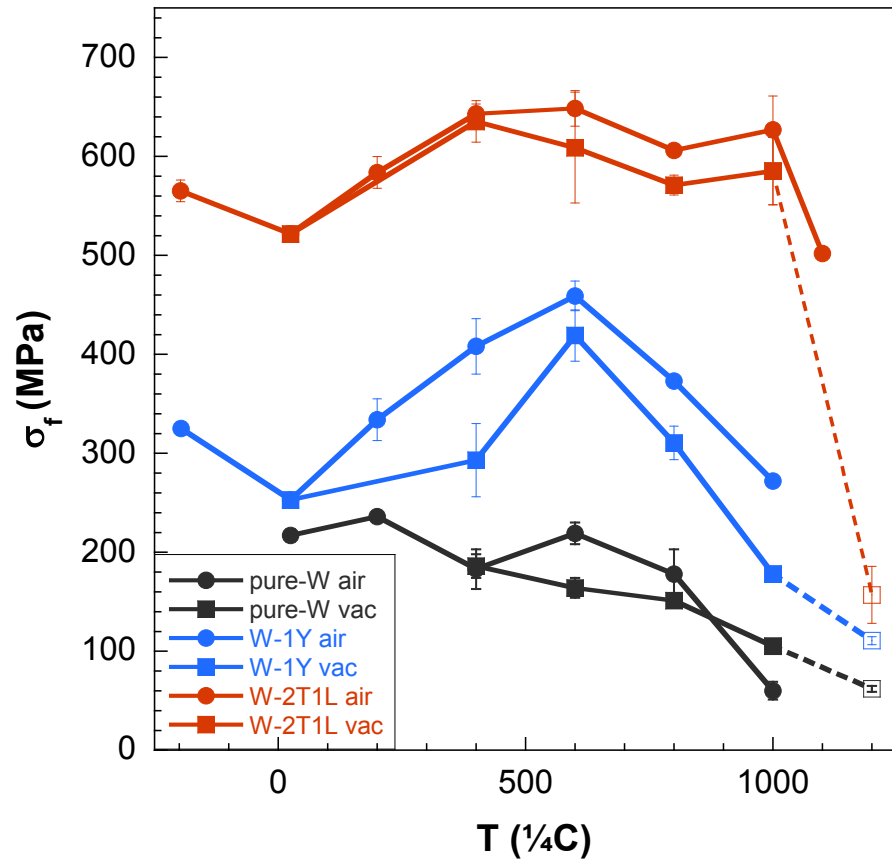
inert atmosphere (vacuum)



$T \leq 1200^\circ \text{C}; 10^{-6} \text{ mbar}$

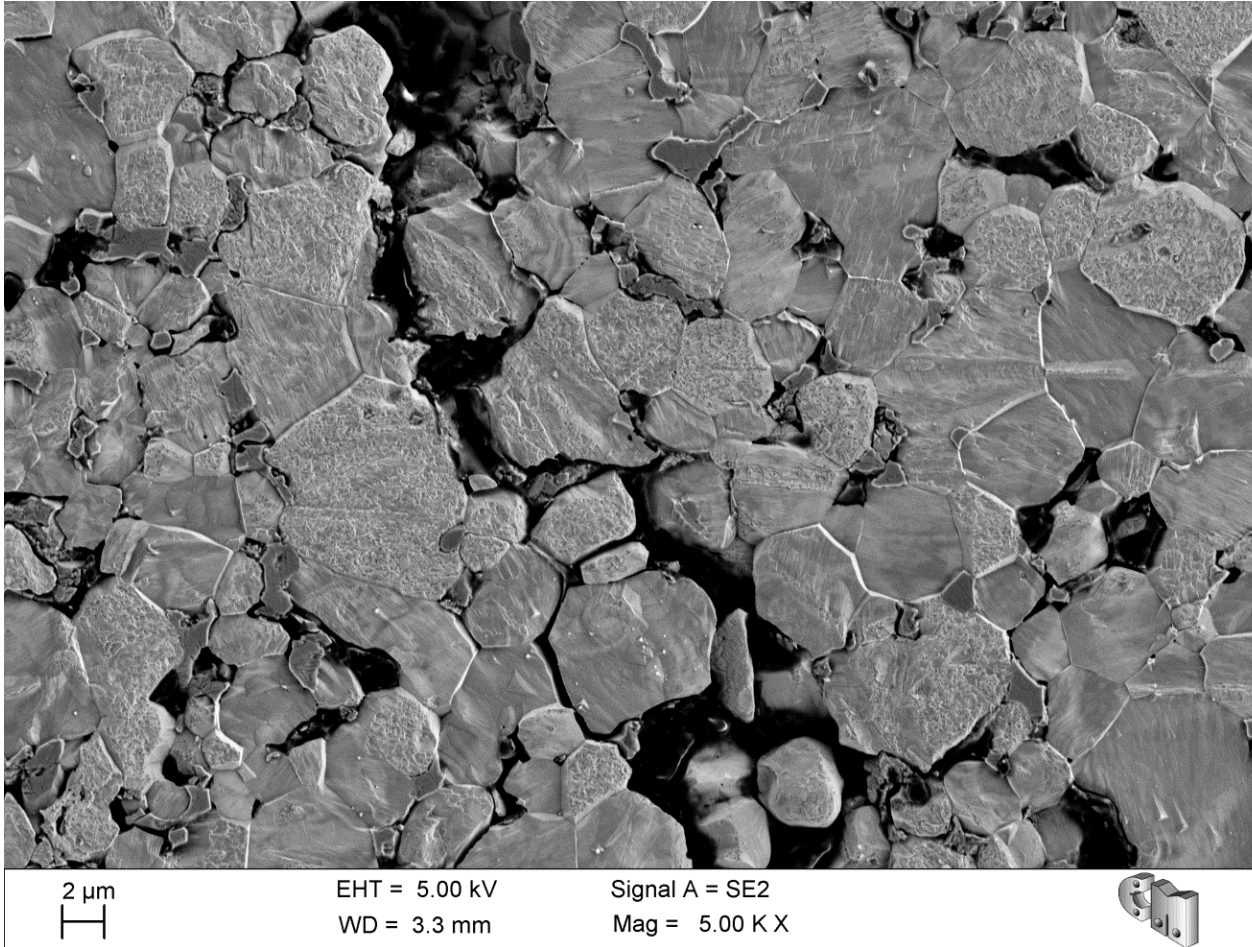


## TPB tests – flexural strength

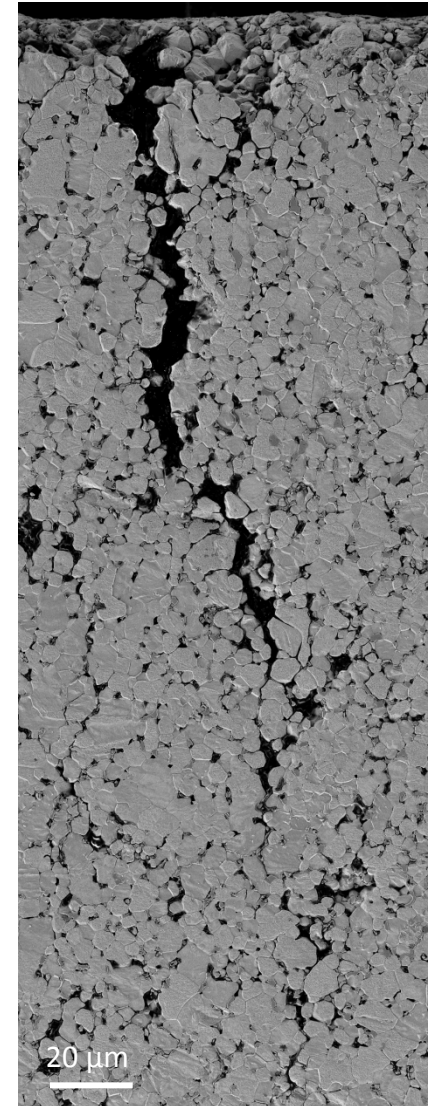


linear elastic until failure (except 1200 ° C)  
 slightly influence of the atmosphere  
 ↑ flexural strength values (W-2T1L)  
 thermal degradation – W-2T1L T>1000 ° C

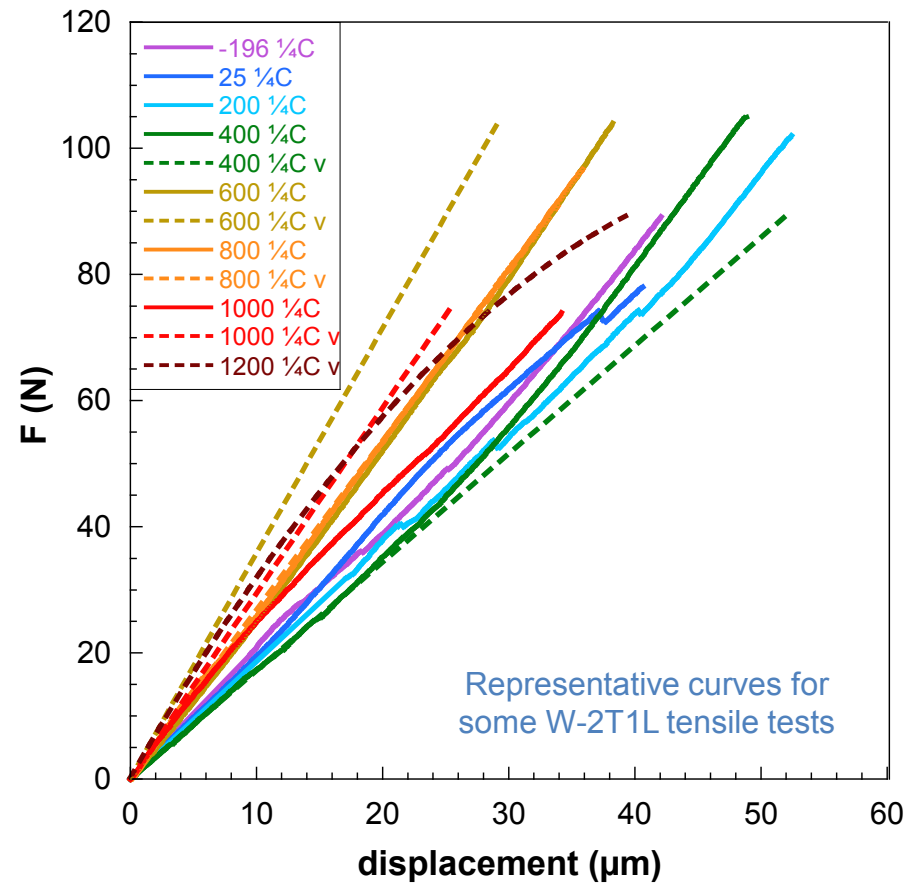
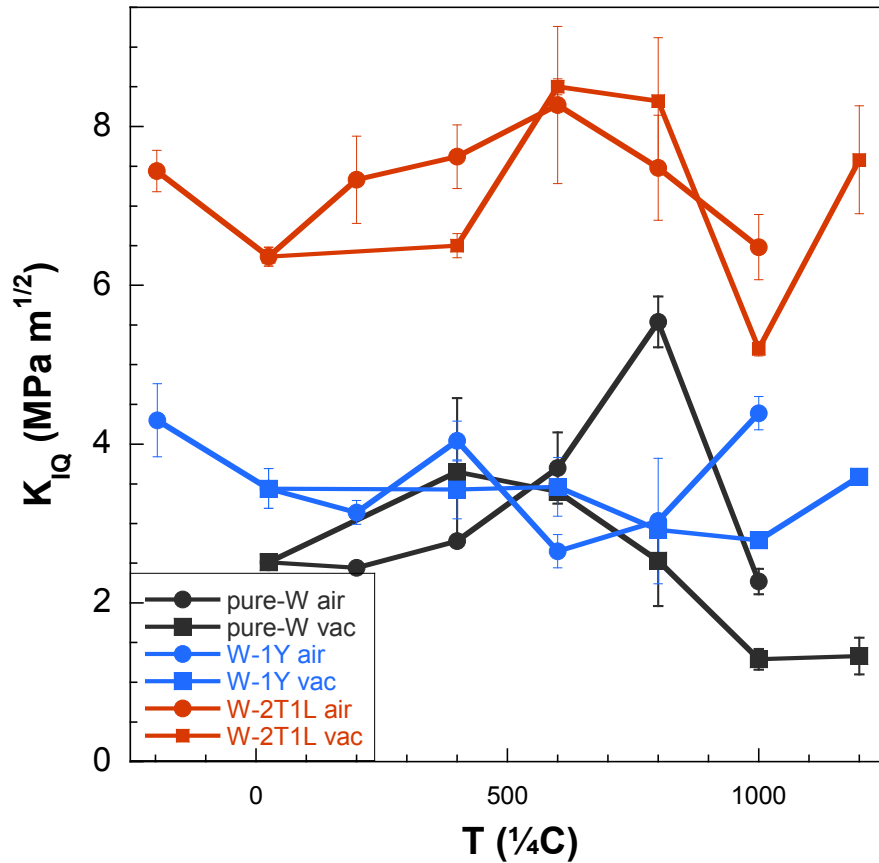
## TPB tests – flexural strength



sample tested a 1200 ° C (not break)  
 intergranular progress of the crack



## TPB tests- fracture toughness



linear elastic until failure (all the T range)  
 improve of the behavior: ↑T (W-1Y), all T (W-2T1L)

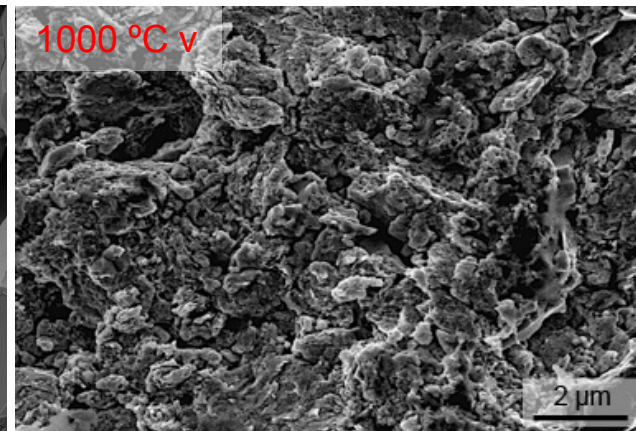
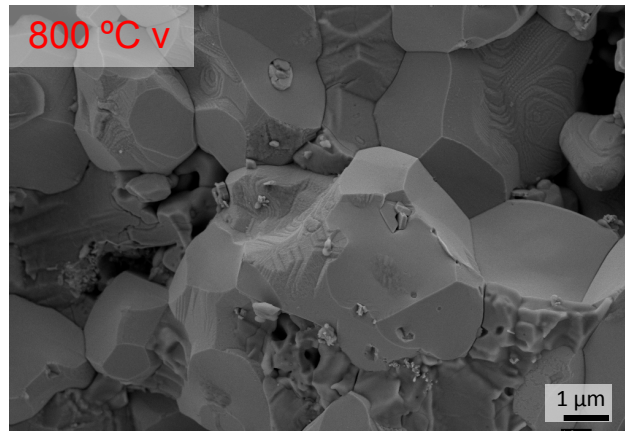
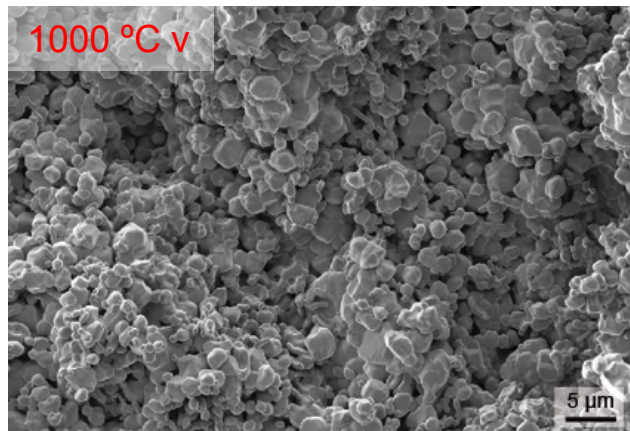
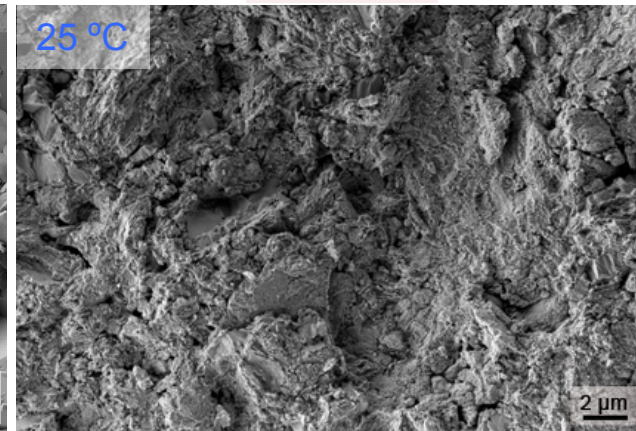
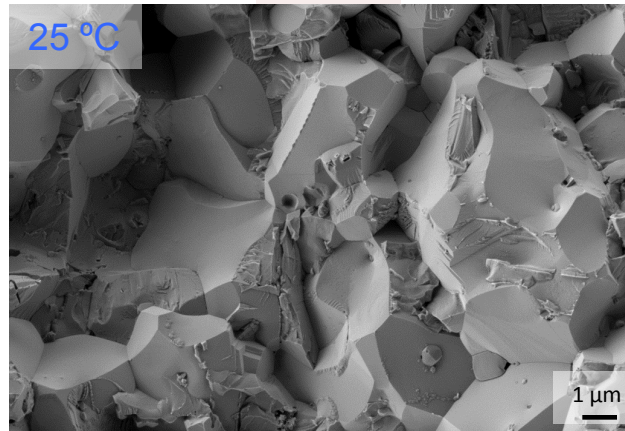
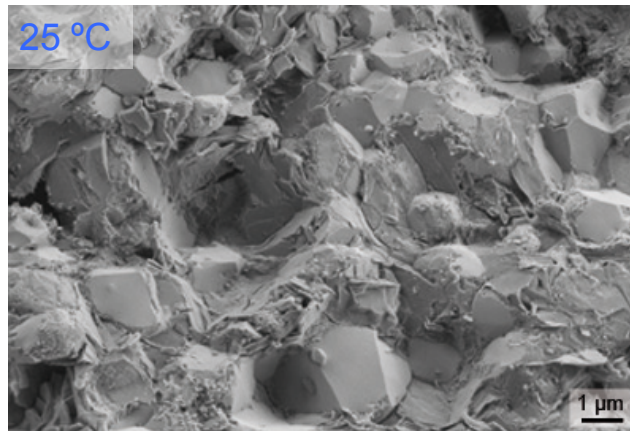


## fracture surfaces

Pure-W

W-1Y

W-2T1L



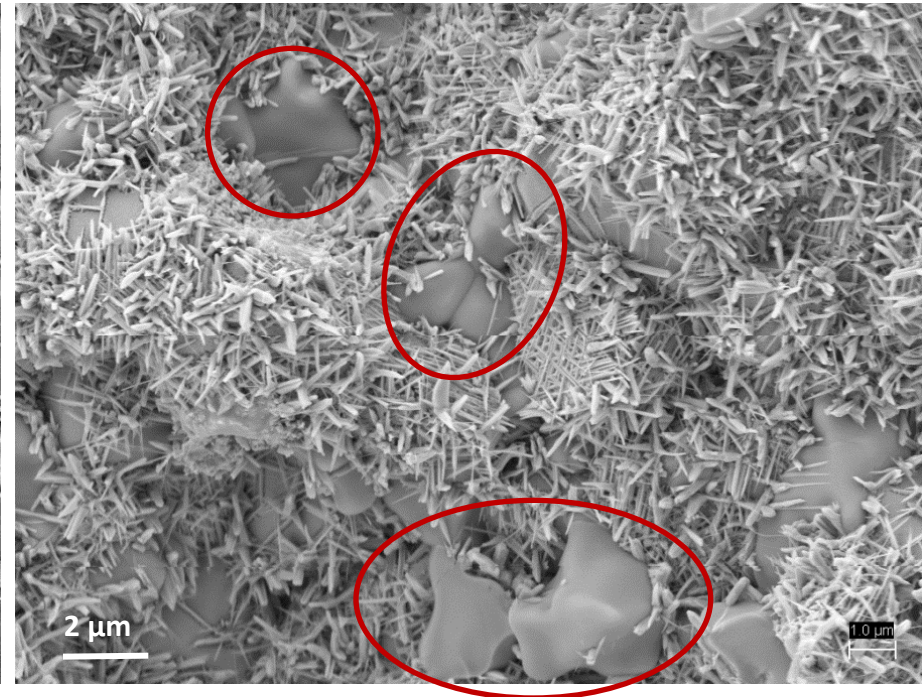
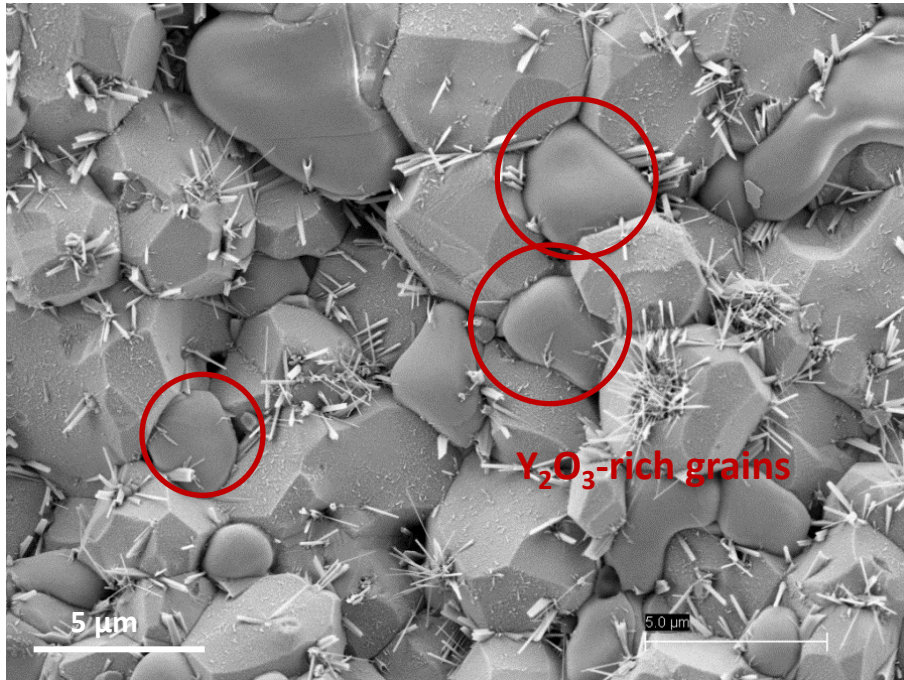
Fracture surfaces of the tested samples at low (up) and high (down) temperature

flat, fracture decohesion by grain boundary



## fracture surfaces

W-1Y 1000 ° C, air



Bimodal morphology of grains:

- polyhedral coarse grains of W (surrounded by oxide needles)
- rounded smaller grains with dispersion of Yttria nanoparticles

## outlook

- Powder route was used to process W-alloys with additions of  $Y_2O_3$ , Ti and  $La_2O_3$ .
- TPB tests were performed to obtain flexural strength, yield strength and fracture toughness (-196 to 1200 ° C, air & vacuum). Moreover nanoindentation, Vickers and IET tests (H, E).
- Laser notching of the samples show a notch like crack with a root radius between 1-50 nm and less dispersion of the results.
- W-1 $Y_2O_3$ :
  - Slightly enhancement of properties at high T, but increased porosity.
  - Microstructure - Yttria nanoparticles located in the grain boundaries and bimodal morphology of grains.
  - Fracture analysis - intergranular breakage between grain boundaries (support the brittle behavior of the performed tests up to high T).
- W-2Ti-1 $La_2O_3$ :
  - The porosity slightly decreases and the grain size became nanostructured.
  - Mechanical properties according to TPB tests increase. The values remain stables in all the temperature range (exc. 1200 ° C, ductile).
  - Fracture surfaces remain flat and show intergranular breakage between grain boundaries. Supported by brittle behavior of the performed TPB tests.

# Thank you for your attention!



POLITÉCNICA



EFDA

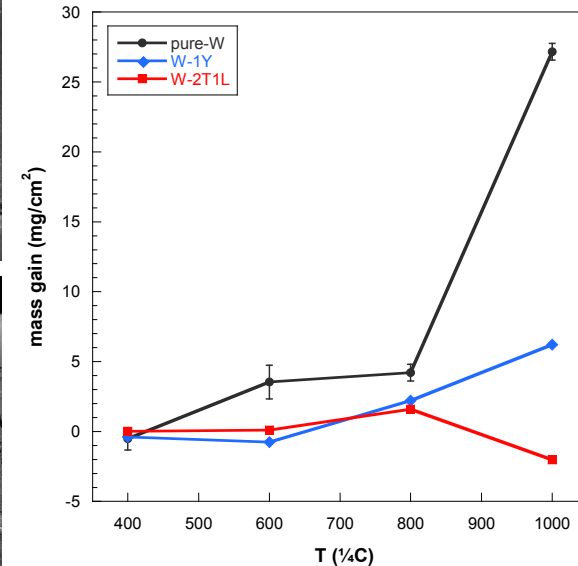
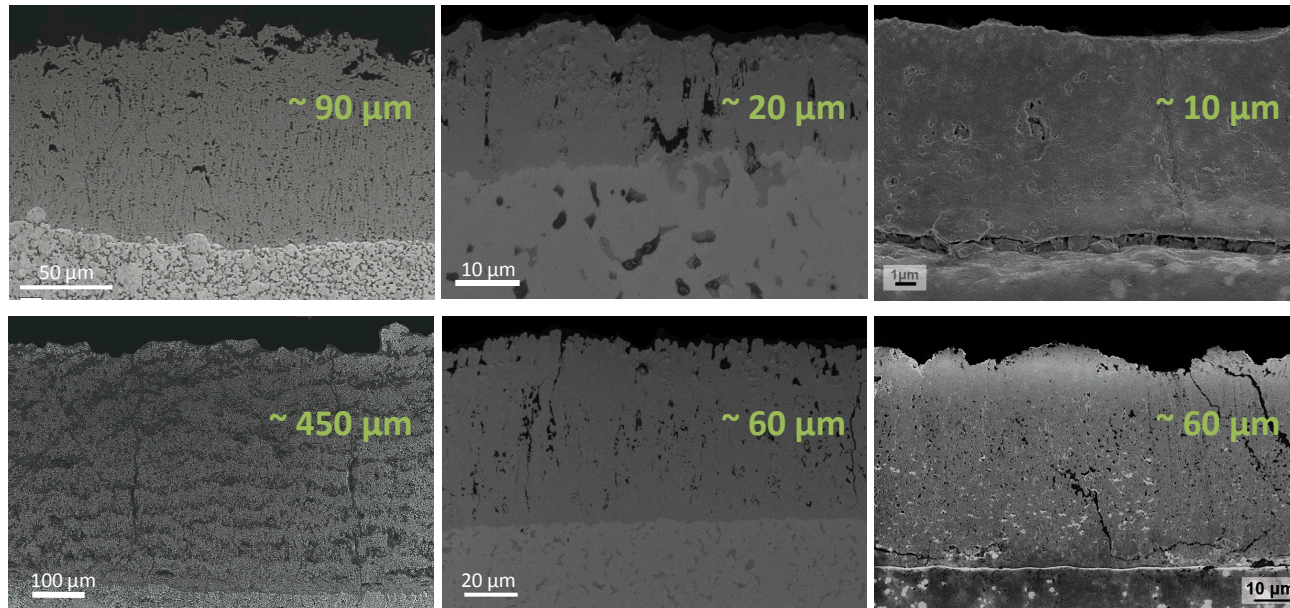


## oxidation

Pure-W

W-1Y

W-2T1L



Cross sections of the scales developed in the samples oxidized in air after 15 min exposure at 800 (up) and 1000 ° C (down)

- Pure-W is highly reactive with oxygen above 400 ° C, but with the addition of the alloying elements (Ti, La<sub>2</sub>O<sub>3</sub>), the oxidation processes decreased (W-1Y, W-2T1L).
- The outer yellow scale remain thinner and the mass gain is lower in all the temperature range
- At 1000 ° C some cracks appears in the scale as a consequence of the stresses produced during oxide growing.